# Improved charge transfer through the minimal addition of Pb as a sintering aid to TiO<sub>2</sub>-based low-temperature dye sensitised solar cell

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Abstract

The poor interparticle connectivity between the nanoparticles architecture of photoanode due to insufficient sintering temperature has been an issue for developing flexible dye sensitised solar cell (DSSC). This issue has led flexible DSSC to yield low conversion efficiency. This research aims to implement lead (Pb) as sintering aid to improve the interparticle connection of the photoanode by using the concept of liquid phase sintering. With low melting point of Pb (327.5°C), necks were formed at the titanium dioxide (TiO<sub>2</sub>)-Pb interface that improved the connection and lowered the electronic resistance even at low sintering temperature of 150-250°C. Morphological studies showed the formation of these necks, while phase analysis indicated the more desirable TiO<sub>2</sub> anatase phase was present. Specimens containing 5wt% Pb in the TiO<sub>2</sub> matrix showed the highest efficiency value of 8.73% at 250°C, which is even higher compared to their high-temperature (450°C) counterpart by 12.21%. This is due to surface fusion of Pb at a lower temperature, leading to enhanced interparticle contact and reduction in recombination reactions. Further increase in Pb did not improve the conversion efficiency which can be due to high charge trapping sites and layer cracking due to high amounts of Pb in TiO<sub>2</sub> matrix.

Keywords: Flexible DSSC; photoanode; TiO<sub>2</sub>; sintering; low temperature

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## **Graphical abstract**



## Highlights

- Lowered resistance and improved connection due to neck formation between TiO<sub>2</sub>-Pb
- Sintering of photoanode at low temperatures (150°C, 200°C and 250°C)
- 12.21% enhancement of DSSC efficiency with Pb sintering aid
- 5wt% Pb in the TiO<sub>2</sub> matrix showed the highest efficiency value of 8.73%

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## **1.0 Introduction**

Dye sensitised solar cells (DSSCs) are a third-generation solar cell that have been growing in popularity due to their low cost of production with readily available materials, easy fabrication process, ability to operate under low-light conditions, good stability, and nice aesthetics [1,2]. DSSC is a sandwich structure comprised of components including the conductive substrate, photoanode, counter electrode, sensitiser/dye and electrolyte. Typically, the conductive substrate is made up of conductive glass materials such as fluorine tin oxide (FTO) glass with one substrate coated with semiconducting materials such as titanium dioxide (TiO<sub>2</sub>) nanoparticles that are sensitised with dyes (ruthenium-based) [3]. Meanwhile, the other substrate is coated with counter electrode catalysts, typically platinum or other materials such as MXene (Ti<sub>3</sub>C<sub>2</sub>) flakes [4] and tungsten disulfide (WS<sub>2</sub>) nanosheets [5]. The two electrodes are then joined together with some iodide-triiodide liquid electrolyte is usually deposited in between them to complete the device. Due to the device's simplicity, DSSC have been under intensive research, with the highest efficiency recorded at 14.2% in the year 2020 [6], a slight increase from the previous record of 12.3% in 2011 [7]. The lack of significant improvement in the efficiency of DSSC highlights their main issue, especially when compared to some other third-generation solar cells, such as perovskite and organic solar cells with conversion efficiency of 25.5% and 19%, respectively [8,9]. Therefore, researchers have been taking a different approach by reducing the cost of DSSC manufacturing instead while still producing good or comparable efficiency.

The cost of DSSC manufacturing can be decreased by developing flexible DSSC using flexible materials such as plastic substrate made up of indium tin oxide-polyethyleneterephtalate (ITO-PET) and indium tin oxide-polyethylenenaphtalate (ITO-PEN). These materials are flexible, lightweight, have good transmittance and conductivity with low production cost by utilising roll-to-roll system to mass produce the substrate in industry [10,11]. The ability of these materials to be shaped easily allowed them to be applicable for mobile and wearable applications, encouraging the development of flexible DSSC even further [12]. Despite all the advantages of the flexible DSSCs, these devices suffer from low conversion efficiency compared to rigid glass-based DSSCs with the highest efficiency is reported at 10.28% [13].

A typical rigid glass-based DSSC underwent sintering at high temperature (usually  $450^{\circ}$ C for TiO<sub>2</sub> photoanode) to improve the interparticle connection between the photoanode nanoparticles, remove organic surfactant in the photoanode paste and enhance the adhesion between the photoanode and substrate [14]. However, this approach is not applicable for

plastic-based DSSC with low thermal stability as the ITO-PET and ITO-PEN polymer substrate
would deform at temperature higher than 150°C [11]. Hence, various research has been
conducted to enhance the efficiency of DSSC at low sintering temperature.
One of the approaches that have been taken to enhance the efficiency of DSSC is doping
the photoanode with metallic cations materials by tuning the band structure of the photoanode
[15]. By adding tin (Sn) as dopant to TiO<sub>2</sub> photoanode, Ni et al. [16] managed to develop DSSC
with good conversion efficiency of 8.75% compared to DSSC with undoped TiO<sub>2</sub> (7.46%).
This was attributed to the Sn dopant that expands the electron mobility and raises the flat band
potential, eventually improved the performance of DSSC. Dubey et al. [17] meanwhile studied
a variety of dopants such as vanadium (V), iron (Fe), chromium (Cr) and cobalt (Co) to TiO<sub>2</sub>
photoanode, which produced DSSC with efficiency of 3.33%, 4.85%, 3.29% and 3.20%

a variety of dopants such as vanadium (V), iron (Fe), chromium (Cr) and cobalt (Co) to TiO<sub>2</sub> photoanode, which produced DSSC with efficiency of 3.33%, 4.85%, 3.29% and 3.20% respectively compared to pure TiO<sub>2</sub> (1.31%). Thus, highlighting the effectiveness and potential in using dopants as there are wide availability of materials that can still be studied. Dopants have also shown to be effective for flexible DSSC that was sintered at low temperature of  $130^{\circ}$ C. This was demonstrated by Song et al. [18] where magnesium (Mg) was used as doping material to TiO<sub>2</sub> photoanode. An increase in efficiency can be seen as Mg-doped TiO<sub>2</sub> DSSC yielded an efficiency of 2.2% compared to undoped ones (1.7%) due to the change in the conduction band state of TiO<sub>2</sub> surface.

By using the surface plasmon resonance effects found in metallic nanoparticles and the concept of liquid phase sintering, another approach can also be taken to improve the DSSC sintering process. Liquid phase sintering utilises at least two chemical elements mixed at different compositions, with the major component remaining as solid while the minor component becomes the aids that forms liquid during sintering [19]. Liquid phase sintering enhances the sintering mechanism by enhancing the diffusion speed and mass transfer mechanism. Ahmad et al. [20] utilises germanium (Ge) as sintering aid to TiO<sub>2</sub> photoanode, where they discovered necks were formed at the TiO<sub>2</sub>-Ge interface during sintering. These necks helped improve the interparticle connection leading to better efficiency from 4.03% for pure TiO<sub>2</sub> photoanode to 5.65% with the addition of Ge sintering aid. In another study, Ahmad et al. [21] applied zinc (Zn) nanoparticles as sintering aid to TiO<sub>2</sub> photoanode sintered at the TiO<sub>2</sub>-Zn interface, improving the efficiency from 4.27% for pure TiO<sub>2</sub> photoanode sintered at 450°C, to 4.92%. Due to the low melting point of Zn (419.5°C), the necks were formed despite the low sintering temperature because of the rules of metallurgical sintering. In metallurgical

sintering, the sintering temperature of metals are typically found at around 67-75% of the metals' melting point [22]. Hence, addition of sintering aid for liquid phase is suitable for both high and low temperature DSSC development.

Lead (Pb) has high durability, density and is a corrosion resistant metal with a low melting point of 327.5°C [23]. Pb is a group IV element similar to Ge and Sn which have shown good photocatalytic activity with TiO<sub>2</sub> has encouraged some previous studies to try reproduce similar outcomes. Pb was used as dopant material with TiO<sub>2</sub> that has tuned the energy bandgap and electronic structure of TiO<sub>2</sub>, extending the wavelength response for solar light, increasing the number of photogenerated electrons and enhances the photocatalytic activity of Pb-doped TiO<sub>2</sub> films [24,25]. Lead sulphide (PbS) nanoparticles with their tuneable band gap have been used as photoanode material for DSSC in a study by Mishra & Saha [26] with power conversion efficiency (PCE) of 1.92%. PbS has also been used with TiO<sub>2</sub> photoanode to form nanocomposite paste in a study by Singh et al. [27] which produced DSSC with efficiency of 0.35%, compared to the pure TiO<sub>2</sub> photoanode (0.08%).

Overall as can be seen from the above literature review, yet extensive studies have not been conducted using pure Pb metal nanoparticles in DSSC development. Besides, the implementation of metal nanoparticles as sintering aid to DSSC photoanode has still not been a widely research topic in this field. However, metal type of nanoparticles have great potential to enhance the power conversion efficiency (PCE) of DSSCs especially for low temperature applications. Thus, this research aims to develop highly efficient DSSC at low sintering temperature with novel implementation of Pb as sintering aid to TiO<sub>2</sub> photoanode. The low melting point and high charge carrier mobility would make them a suitable sintering aid to form necks with the  $TiO_2$  nanoparticles despite the low sintering temperature [23]. Several different compositions of Pb and low sintering temperatures are investigated to study their effects on the performance of DSSC as well as determining the optimal condition. Highest PCE (8.73%) is found to be for specimens containing 5wt% Pb in the TiO<sub>2</sub> matrix at 250°C which outperform as compared to their high-temperature (450°C) counterpart (having PCE of 7.78% ) by 12.21%. Besides, the low cost of Pb due to the abundance of the material is another key factor in supporting the viability of this novel research to develop low cost DSSC with good efficiency.

## 2.1 Materials

Titania (TiO<sub>2</sub>) paste (Sigma-Aldrich), fluorine tin oxide (FTO) glass (Sigma-Aldrich, surface resistivity:  $13\Omega/sq$ ), Pb powder (Sigma-Aldrich, 100 mesh), platinum (Pt)-coated FTO glass (Solaronix), iodolyte An-50 (Solaronix), ruthenizer 535-bisTBA (Solaronix), methanol (Fisher Scientific), acetonitrile (Fisher Scientific), ethanol (Fisher Scientific), binder clips.

## 2.2 Preparation of Pb sintering aid

20g of Pb powder (100 mesh) was deposited into a grinding bowl with zirconia balls for the grinding process using a planetary ball mill (Fritsch Pulverisette 5) operated at 500 rpm. Multiple grinding cycles were conducted until the Pb powder was found to be in nanoparticle size of around 450-500 nm, checked by using a particle size analyser (Anton Paar Lifesizer 500).

## 2.3 Preparation of photoanode samples

Pure FTO glass was cleaned using methanol each time before they were deposited with photoanode materials. A coating area of 1cm x 1cm was prepared by taping up some parts of the FTO glass using cellophane tape. Pure TiO<sub>2</sub> photoanode samples were prepared by depositing some TiO<sub>2</sub> paste onto the coating area and was coated via doctor blade method. The coated sample was left to dry for 1 hour before the tape was removed, leaving a smooth coated surface. The coated FTO glass was then placed onto a hotplate (IKA RCT basic) and was heated at several different temperature (150°C, 200°C, 250°C and 450°C) for 3 hours. After the sintering process, the samples were then kept and secured for the next steps.

In order to prepare the Pb-TiO<sub>2</sub> photoanode samples, different amount of grinded Pb powder was added to the TiO<sub>2</sub> paste and mixed to form different weight composition of Pb (0.5wt%, 1wt%, 3wt%, 5wt% and 10wt%). Similar to the previous step, the doctor blade method was used to coat the composite paste onto the FTO glass to form a coating area of 1cm x 1cm. The samples were then heated at different temperatures ( $150^{\circ}C$ ,  $200^{\circ}C$  and  $250^{\circ}C$ ) for 3 hours. The sintered photoanode samples were then kept and secured for further use.

## 2.4 Fabrication of DSSC devices

Ruthenium-based dye solution was prepared by dissolving 10mg of ruthenizer powder into 5ml of acetonitrile solution. The photoanode sample was soaked into the dye solution and left inside for 1 day. After 24 hours, the photoanode sample was taken out from the dye solution and was left to dry at room temperature for 5 hours. Pt-coated FTO glass was then combined together with the dyed photoanode sample. A drop of electrolyte was deposited into the sample via a small hole on the Pt-coated FTO glass to complete the device. Binder clips were then used to clip the two components together. Similar process was repeated for all the photoanode samples and the devices were then kept for further use.

## 2.5 Characterisation and analysis process

Several characterisation studies were conducted on the photoanode samples using equipment such as the field emission scanning electron microscope (FESEM model: Hitachi SU8010) and transmission electron microscope (TEM model: FEI Tecnai G2 20 S-TWIN) for morphological studies. Elemental composition and mapping of the photoanode samples were also investigated using the scanning electron microscope-energy dispersive x-ray (SEM-EDX model: TESCAN VEGA 3) analysis. Phase analysis was also conducted using x-ray diffraction (XRD) analysis (XRD model: Rigaku Miniflex 600). Light absorption studies were also conducted using UV-visible spectroscopy (UV-vis model: PerkinElmer Lambda 750). Electrochemical impedance spectroscopy (EIS) analysis was also conducted using Gamry Interface 1010E Potentiostat to plot the Nyquist and Bode plot. Nyquist plot was used to check for the series (R<sub>S</sub>) and charge transfer resistance (R<sub>CT</sub>) while the Bode plot was applied to determine the electron lifetime (t) of the DSSC devices. The frequency range was set at 0.1 Hz - 1 MHz and the voltage were set at 10 mV for all samples. Solar simulator (solar simulator model: IVT 300) was also used to measure the current-voltage (I-V) curves, short circuit current (J<sub>SC</sub>), open circuit voltage (V<sub>OC</sub>), fill factor (FF) and power conversion efficiency (PCE). A 1600W Xenon lamp (SAN-EI electric Co., Ltd) was used in IVT 300 series solar simulator. The caliberation was done before testing using a certified golden solar cell (Amorphous Silicon solar cell). The reason behind using a standard Amorphous Si solar cell it was due to minimum mismatch. The mismatch was found to be approximately 3.7% and for proper measurements it was adjusted in machine parameters as a mismatch factor. The data obtained were then analysed and tabulated accordingly in the research. At least 5 samples for each selected sample have been tested and the average value has been presented in this study.

## 3.0 Results and discussion

## 3.1 Morphological studies

FESEM equipment have been used to capture the image of the prepared photoanode samples to analyse their morphologies at different sintering temperatures and with the addition

of Pb sintering aids. Figure 1 shows the coated and sintered photoanode samples on top of FTO glasses were used for the FESEM analysis as well as the elemental mapping and composition via the SEM-EDX analysis.



Figure 1: a) FESEM image of pure TiO<sub>2</sub> photoanode sintered at 150°C b) FESEM image of pure TiO<sub>2</sub> photoanode sintered at 250°C c) FESEM image of TiO<sub>2</sub>-5% Pb photoanode sintered at 200°C d) FESEM image of TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C e) Schematic diagram of the necking between the TiO<sub>2</sub>-Pb nanoparticles f)-i) SEM-EDX elemental mapping for Ti, O and Pb elements respectively j) EDX report for TiO<sub>2</sub>-5% Pb photoanode sample

FESEM images of the prepared photoanode samples have been depicted in Figure 1a-d. The effect of sintering temperature can be observed from the FESEM images, where the porosity of the films seems to have increased as the sintering temperature increased. The TiO<sub>2</sub>

nanoparticles in Figure 1b have also agglomerated more with a denser arrangement than the nanoparticles in Figure 1a due to the increase in sintering temperature. The increase in sintering temperature has enlarged the surface area for better dye absorption, light harvesting and eventually DSSC performance.

Meanwhile, Figures 1c and 1d highlight the presence of Pb sintering aids in the photoanode films. The TiO<sub>2</sub> nanoparticles in both figures have shown better connectivity due to the presence of Pb sintering aid as indicated in the figure. The cause for this improved connection can be attributed to the aforementioned liquid phase sintering and metallurgical sintering theory. Pb nanoparticles have been melted during the sintering process due to their low melting point of 327.5°C. Since the sintering temperature of metal nanoparticles is typically around 70% of their melting point, Pb nanoparticles were sintered well at 250°C [22]. The melting of Pb nanoparticles then caused the formation of necks at the TiO<sub>2</sub>-Pb interface as shown in Figure 1e, which would enhance the interparticle connectivity of the photoanode. The presence of Pb nanoparticles with the desired elemental composition was also observed in the EDX report of the TiO<sub>2</sub>-5% Pb photoanode sample. Elemental mapping for the aforementioned photoanode sample has also been conducted showing the presence of all the elements required in the formation of necks at the TiO<sub>2</sub>-Pb interface. The absence of other elements in the EDX analysis indicates that there were no traces of impurities in the prepared photoanode samples. Hence, highlighting the preparation of photoanode samples have been prepared accordingly in the research.

Besides FESEM analysis, TEM analysis was also conducted and their images were also observed to study the morphologies of the photoanode samples prepared as shown in Figure 2.



Figure 2: TEM images of photoanode samples a) pure TiO<sub>2</sub> sintered at 250°C b) TiO<sub>2</sub>-5% Pb sintered at 250°C with lower magnification c) TiO<sub>2</sub>-5% Pb sintered at 250°C with higher magnification

Figure 2 showed the TEM images of photoanode samples prepared at 250°C with and without the addition of Pb sintering aid. The effect of the implementation of Pb sintering aid can clearly be seen in Figure 2b & 2c where there has been an increase in the aggregation between the nanoparticles with the formation of larger chunks of nanoparticles. The highlighted section in Figure 2b & 2c indicates the formation of necks between the TiO<sub>2</sub>-Pb nanoparticles during the sintering process. The presence of these nanoparticle aggregations in the TEM images further supports the observation made in the FESEM images regarding the formation of necks that improve the interparticle connection.

Phase identification of photoanode samples has been conducted via the XRD analysis as shown in Figure 3.



Figure 3: a) XRD analysis of pure TiO<sub>2</sub> photoanode samples at multiple sintering temperature b) XRD analysis of TiO<sub>2</sub> - 5%Pb photoanode samples at multiple sintering temperature with unit cells of anatase TiO<sub>2</sub> and rutile TiO<sub>2</sub> [28]

Figure 3a indicates that the peaks have been formed at similar areas for each photoanode sample even with different sintering temperatures. This observation shows that the crystallite structure of  $TiO_2$  did not experience any phase change despite the different sintering temperatures. The more desirable  $TiO_2$  anatase phase have been found to be more prevalent

than the less desirable rutile phase as highlighted in the peaks shown in Figure 3a. Peaks were formed at 20 positions of 26°, 39°, 49°, 56° and 64° with miller indices of (101), (004), (200), (211) and (204) planes, respectively, indicating TiO<sub>2</sub> anatase phase [29]. Meanwhile, a few peaks formed at 27° and 36° with miller indices of (110) and (200) planes, respectively which indicated the TiO<sub>2</sub> rutile phase. The unit cells showed that the Ti-Ti distance is longer, while the Ti-O distance is shorter in anatase phase compared to the rutile phase. This structure would then cause a difference in the mass densities and electronic band structures [30]. Hence, anatase TiO<sub>2</sub> have larger bandgap, higher surface energy, lower recombination reaction and smaller grain size, for better photocatalytic and photovoltaic activity [29,31]. These advantages would eventually lead to a higher DSSC efficiency, making TiO<sub>2</sub> anatase phase more favorable. Pb sintering aid has also been found to not have caused a change in the TiO<sub>2</sub> phase. TiO<sub>2</sub> anatase have remained dominant despite adding Pb as indicated by the peaks formed at 32° with miller indices (111) plane as shown in Figure 3b [32]. From these observations, it can be deemed that the low sintering temperature of 150-250°C as well as with the addition of Pb nanoparticles as sintering aid were suitable for maintaining good TiO<sub>2</sub> anatase phase.

## **3.2 Light absorption studies**

Light absorption studies for the photoanode materials have been conducted by dissolving an appropriate amount of TiO<sub>2</sub>, Pb, and TiO<sub>2</sub>-5% Pb into ethanol solvent before they were sonicated for 30 minutes. Pure ethanol solution was then used as a reference, and the outcome from the UV-visible spectroscopy was shown in Figure 4. From the UV-vis spectra, the absorbance peak for TiO<sub>2</sub> sample was at the wavelength of 350 nm. This falls under the UV region as shown in Figure 4, typical for TiO<sub>2</sub> nanoparticles [33]. Similar finding was also obtained for the TiO<sub>2</sub>-5% Pb sample where the absorbance peak was formed at a wavelength of 340 nm, also falling in the UV region. The presence of Pb particle can also be found in the nanocomposite with the formation of peak at around 270 nm, similar to the Pb sample. Apart from that, it can be seen that there was higher absorbance for the nanocomposite sample as opposed to pure TiO<sub>2</sub> and Pb sample. The addition of Pb nanoparticles was observed to have helped increase the absorbance of TiO<sub>2</sub>-5% Pb sample even beyond the UV region. This was proven by the absorbance peak has been formed at the near-infrared region with wavelength of 870 nm, similar to the peak formed by the pure Pb nanoparticle as shown in Figure 4. Hence, indicating the impact of adding Pb nanoparticles as sintering aid to add wavelength response in the near-infrared region for the TiO<sub>2</sub>-Pb nanocomposite sample. The improvement in the nanocomposite sample to absorb light at both UV and near-infrared region would help boost

their light absorption capabilities. This would eventually enhance the conversion efficiency of the solar device and encourage the implementation of Pb as sintering aid.



Figure 4: UV-visible spectra of Pb, TiO<sub>2</sub> and TiO<sub>2</sub>-5% Pb in ethanol solution

## 3.3 Electron transfer studies

Figure 5 and 6 depicts the Nyquist and Bode plot from EIS analysis which was performed to study the electron transfer of the complete DSSC devices. EIS response for DSSC devices prepared at 250°C sintering temperature for different Pb composition have been studied. In theory, 250°C was determined to be the most suitable sintering temperature that would yield good results by considering the metallurgical sintering concept [22]. The effect of Pb composition was then studied by analysing the EIS response for these DSSC devices.



Figure 5: a) Nyquist plot for DSSC devices sintered at 250°C with equivalent circuit b) Bode plot for DSSC devices sintered at 250°C c) Schematic diagram of electron pathway between pure TiO<sub>2</sub> photoanode samples and TiO<sub>2</sub>-Pb composite photoanode sample

From the Nyquist plot, the charge transfer resistance ( $R_{CT}$ ) of DSSC devices was observed to have decreased as Pb sintering aid was implemented to TiO<sub>2</sub> photoanode. The  $R_{CT}$  for DSSC based on pure TiO<sub>2</sub> photoanode was found to be 22.572  $\Omega/cm^2$ , while the  $R_{CT}$  for DSSC based on TiO<sub>2</sub>-0.5% Pb was 18.317  $\Omega/cm^2$ . The result indicated that the charge transfer capability of the composite photoanode was better. The improvement was attributed to the neck formation between the TiO<sub>2</sub>-Pb nanoparticles during the sintering process. The formation of these necks increased the interparticle connectivity from the surface fusion of Pb and the increase in electron density. This occurrence would then led to shorter electron pathway for the transfer of electrons throughout the nanoparticles. With shorter electron pathway, the charge transfer would be improved and the charge transfer properties of the TiO<sub>2</sub>-Pb photoanode has been enhanced. Thus, the recombination reactions and resistance of DSSC devices have been reduced by adding Pb sintering aid as opposed to pure TiO<sub>2</sub> photoanode with higher resistance values. The  $R_{CT}$  was reduced even further as the composition of Pb was increased. The lowest  $R_{CT}$  value was discovered when TiO<sub>2</sub>-5% Pb photoanode was applied with a value of 11.54

  $\Omega$ /cm<sup>2</sup>. However, the resistance was found to have increased when the composition of Pb was increased to 10% with high R<sub>CT</sub> value of 49.29  $\Omega$ /cm<sup>2</sup>, higher than pure TiO<sub>2</sub> photoanode. This occurrence was due to excessive amount of Pb, which became impurities that causes electron trapping sites and surface defects [21].

The results from the Nyquist plot were supported even further with the study of Bode plot. From Bode plot, the electron lifetime (t) of the DSSC devices was determined using Equation (1) [17]:

$$t(ms) = \frac{1}{2\pi f_{max}} \tag{1}$$

Where,  $f_{max}$  is the maximum frequency from the Bode plot.

The *t* value for DSSC based on pure TiO<sub>2</sub> photoanode was calculated to be 179.229 ms while the *t* value for DSSC based on TiO<sub>2</sub>-5% Pb photoanode was calculated to be 358.861 ms. The longer the electron lifetime, the lower the possibility of recombination reactions to occur at the TiO<sub>2</sub> network. With low charge transfer resistance, longer electron lifetime and less possibilities of recombination reaction, TiO<sub>2</sub>-5% Pb has been concluded as the most suitable composition for DSSC development. Meanwhile, TiO<sub>2</sub>-10% Pb composition was considered unsuitable and worse than pure TiO<sub>2</sub> due to the high R<sub>CT</sub> and low *t* (25.887 ms), with high recombination reaction probability. Further analysis of DSSC fabricated using the TiO<sub>2</sub>-5% Pb composition was then conducted to study the effect of sintering temperature.



Figure 6: a) Nyquist plot for various DSSC devices with equivalent circuit b) Bode plot for various DSSC devices

EIS analysis for DSSC devices based on TiO<sub>2</sub>-5% Pb composition at different sintering temperature was conducted and compared with pure TiO<sub>2</sub> DSSC devices. Figure 6a showed that  $R_{CT}$  values decreased as the sintering temperature increases. The highest  $R_{CT}$  value was found for DSSC based on pure TiO<sub>2</sub> sintered at 150°C with a value of 72.966  $\Omega/cm^2$ . The photoanode's smaller porosity and surface area lowered the electrical contact between nanoparticles with low electron transfer that increased the resistance and recombination reaction [34]. To prevent such issue, TiO<sub>2</sub> photoanode is typically sintered at 450°C [35]. As shown in Figure 6a, the pure TiO<sub>2</sub> photoanode sintered at 450°C yielded lower  $R_{CT}$  value of 14.883  $\Omega/cm^2$ , highlighting the importance of having high sintering temperature. However, in this research, the photoanode samples with Pb addition were sintered at low temperature of 150°C-250°C to be applicable for flexible DSSC development [36].

Additionally, the absence of sintering aid also hindered the connection between the TiO<sub>2</sub> nanoparticles. Lower R<sub>CT</sub> values (22.418  $\Omega$ /cm<sup>2</sup>) was obtained when 5% Pb was added, even at 150°C sintering temperature. The lowest R<sub>CT</sub> value of 11.54  $\Omega$ /cm<sup>2</sup> was obtained when TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C was implemented. The value was even lower than

the control DSSC based on TiO<sub>2</sub> photoanode sintered at 450°C ( $R_{CT} = 14.883 \ \Omega/cm^2$ ). Due to the concept of liquid phase and metallurgical sintering, Pb, with a melting point of 327.5°C, has melted and formed necks at the TiO<sub>2</sub>-Pb interface [19]. The interparticle connection of the photoanode have been greatly enhanced, while the charge transfer resistance and recombination reactions have been lowered. Besides the Nyquist plot, Bode plot have also been analysed to calculate the electron lifetime of DSSC. The longest electron lifetime value was obtained for DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C with a value of 358.861 ms. Thus, lowering the recombination reactions of the DSSC and lowering the resistance as previously mentioned. From the EIS analysis, DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C has demonstrated the potential to have the best performance.

## 3.4 I-V curve

The current-voltage (I-V) characteristics of the prepared DSSC devices have been determined using a solar simulator and was plotted in Figure 7. From the solar cell I-V curve, several important parameters in evaluating the performance of DSSC have also been calculated and tabulated in Table 1. Dye loading for the DSSC was also performed via the absorption-desorption method, and sodium hydroxide (NaOH)-ethanol solution was used to desorb the dye loaded on the photoanode samples. The outcome of the dye loading was also presented in Table

1. Control DSSC device prepared with pure  $TiO_2$  photoanode sintered at 450°C was used as reference, thus, comparisons can be made between the prepared and control DSSC.



Figure 7: I-V curve for the prepared DSSC devices

Increased sintering temperature burned out more organic compounds and impurities in  $TiO_2$  nanoparticles, which led to an enlarged surface area [34]. This then improved their dye loading capability, eventually enhancing their light harvesting capability. Meanwhile, improvement in the interparticle connection between the  $TiO_2$  nanoparticles from the addition of Pb sintering aid have also shown to have improved the dye loading ability of the photoanode samples. The highest dye loading was obtained for the  $TiO_2$ -5% Pb photoanode sample sintered at 250°C with a value of  $1.17 \times 10^{-7}/cm^2$ , higher than the control DSSC sample. Thus, indicating that the prepared nanocomposite photoanode have good dye loading and light absorption capability that would eventually enhance the DSSC performance.

 Table 1: Summary of parameters of DSSC samples

Photoanode sample	Sintering temperature (°C)	PCE (%)	Voc (V)	J <sub>SC</sub> (mA/cm²)	FF (%)	Dye loading (× 10 <sup>-7</sup> /cm <sup>2</sup> )
TiO <sub>2</sub>	150	2.70	0.59	6.13	56	1.01

TiO <sub>2</sub>	200	3.87	0.61	9.34	63	1.03
TiO <sub>2</sub>	250	4.48	0.60	11.14	71	1.03
TiO <sub>2</sub>	450	7.78	0.61	16.93	74	1.06
TiO2 - 5% Pb	150	4.17	0.60	10.58	71	1.02
TiO <sub>2</sub> - 5% Pb	200	6.86	0.63	13.64	72	1.08
TiO2 - 5% Pb	250	8.73	0.64	17.78	74	1.17
TiO2-5% Pb	450	3.03	0.60	7.36	64	n/a

Table 1 shows that the open circuit voltage (Voc) values for the samples underwent minimal changes and were slightly increased when Pb was added as a sintering aid. The values of Voc depend on the difference between the Fermi energy level of photoanode and electrolyte's redox potential [37]. The short circuit current (Jsc) was discovered to have improved when the sintering temperature was increased and Pb sintering aid was added. The increased J<sub>SC</sub> can be attributed to the enhanced dye absorption with higher charge transfer and lower recombination in the photoanode [15,37]. The increased interparticle contact, porosity and enlarged surface area from the increase in sintering temperature, have caused these changes, with high  $J_{SC}$  value of 16.93 mA/cm<sup>2</sup> obtained for DSSC sintered at 450°C. Meanwhile, the lowest  $J_{SC}$  (6.13 mA/cm<sup>2</sup>) was yielded by DSSC photoanode sintered at a low temperature of 150°C with a small surface area and low dye absorption. DSSC based on TiO2-5% Pb photoanode sintered at a lower temperature of 250°C has also yielded the highest J<sub>SC</sub> value. Thus, indicating that the presence of Pb nanoparticles has also improved the interparticle connectivity, dye absorption, and charge transfer. The recombination reaction has also been lowered as discussed in the EIS analysis. The fill factor (FF) of DSSCs was increased when the charge transfer resistance was reduced. The highest FF value was 74%, yielded by the two DSSC devices with the lowest charge transfer resistance which were the TiO<sub>2</sub>-5% Pb sintered at 250°C and the control DSSC sample sintered at 450°C, as discussed in the EIS analysis.



value also been increased. The highest PCE value obtained in the research was 8.73% for DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C. This value was even higher than control DSSC prepared with pure TiO<sub>2</sub> at high sintering temperature of 450°C with PCE of 7.78%. Hence, highlighting the superiority in performance of the TiO<sub>2</sub>-Pb photoanode despite the low sintering temperature. The improved interparticle connection due to the necking at the TiO<sub>2</sub>-Pb interface during the liquid phase sintering process have been attributed as the main cause for such good performance. The yielded PCE value have also been supported by the low R<sub>CT</sub> value obtained in the EIS analysis. With lower R<sub>CT</sub>, the electrons were easily transferred into the TiO<sub>2</sub> surface, enhancing the DSSC performance. The nanocomposite photoanode have also shown the highest dye loading value of  $1.17 \times 10^{-7}$ /cm<sup>2</sup>, indicating high light harvesting capability. Besides, DSSC sample based on TiO2-5% Pb photoanode sintered at high temperature of 450°C has also been prepared for comparison purposes with the low temperature DSSC. The result showed that the PCE value for this sample to be lower at 3.03%. The decrease

TiO<sub>2</sub>-5% Pb

in efficiency could be attributed to the oxidation of Pb in the TiO<sub>2</sub> matrix since the oxidation temperature of Pb is around 450°C-640°C [38]. Apart from that, the high sintering temperature has caused Pb to be completely melted and causing excessive liquid phase formation that lowers the porosity of the photoanode [39]. Thus, lowering the dye absorption capability of the photoanode and eventually lowers the performance of DSSC. Hence, the DSSC based on TiO<sub>2</sub>-5% Pb @ 250°C photoanode has the best performance that was even better than the control DSSC device. These results in Figure 8 highlighted the advantage of liquid phase sintering with Pb sintering aid addition method in yielding high performing DSSC even at low sintering temperature.

## 4.0 Conclusions

The research aims to improve the efficiency of DSSC devices being prepared at relatively low sintering temperature. Metallic Pb particles as sintering aid has been employed to reduce the sintering temperature of  $TiO_2$  based photoanode. Various concentrations of Pb from 0.5-10wt% has been tried at different sintering temperatures to optimise the electrooptical properties. Increase in Pb concentration and temperature leads to improved PCE. This is due to improved interparticle contact due to surface fusion of Pb. DSSC prepared with TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C exhibited highest PCE value of 8.73% surpassing even the high temperature DSSC control samples by approximately 12.21% prepared with pure TiO<sub>2</sub> photoanode sintered at 450°C (PCE: 7.78%). The addition of Pb as sintering aid improved the interparticle connection leading to increased in current density and a reduction in recombination reactions. Further increase in concentration of Pb did not improve the PCE which may be correlated with increased impurities in the  $TiO_2$  matrix, which in turn leads to layer cracking and increased charge trapping sites. Besides Pb, other metallic nanoparticles with low melting point should also be researched in the future by using the liquid phase sintering applied in this research. Metal nanoparticles with a lower melting point could produce highly efficient DSSC at lower sintering temperatures that would be suitable for flexible DSSC substrate. The method has shown to be effective for low sintering temperature operation, is a simple process and there is wide availability of metal nanoparticles that are potentially suitable to be used for future research.

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## **Declaration-of-interests**

The authors declare that they have no known competing financial interests or personal relationships that could have influenced the work reported in this paper.

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## Improved charge transfer through the minimal addition of Pb as a sintering aid to TiO<sub>2</sub>-based low-temperature dye sensitised solar cell

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#### Abstract

The poor interparticle connectivity between the nanoparticles architecture of photoanode due to the insufficient sintering temperature has been an issue for developing flexible dye sensitised solar cell (DSSC). This issue has led flexible DSSC to yield low conversion efficiency. This research aims to implement lead (Pb) as sintering aid to improve the interparticle connection of the photoanode by using the concept of liquid phase sintering. With low melting point of Pb (327.5°C), necks were formed at the titanium dioxide (TiO<sub>2</sub>)-Pb interface that improved the connection and lowered the electronic resistance even at low sintering temperature of 150-250°C. Morphological studies showed the formation of these necks, while phase analysis indicated the more desirable TiO<sub>2</sub> anatase phase was present. Specimens containing 5wt% Pb in the TiO<sub>2</sub> matrix showed the highest efficiency value of 8.73% at 250°C, which is even higher compared to their high-temperature, leading to enhanced inter-particle interparticle contact and reduction in recombination reactions. Further increase in Pb did not improve the conversion efficiency which can be due to high charge trapping sites and layer cracking due to high amounts of Pb in TiO<sub>2</sub> matrix.

Keywords: Flexible DSSC; photoanode; TiO<sub>2</sub>; sintering; low temperature

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**Graphical abstract** 



## Highlights

- Lowered resistance and improved connection due to neck formation between  $TiO_2$ -Pb
- Sintering of photoanode at low temperatures (150°C, 200°C and 250°C)
- 12.21% enhancement of DSSC efficiency with Pb sintering aid
- 5wt% Pb in the TiO<sub>2</sub> matrix showed the highest efficiency value of 8.73%

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#### **1.0 Introduction**

Dye sensitised solar cells (DSSCs) are a third-generation solar cell that have been growing in popularity due to their low cost of production with readily available materials, easy fabrication process, ability to operate under low-light conditions, good stability, and nice aesthetics [1,2][1,2]. DSSC is a sandwich structure comprised of components including the conductive substrate, photoanode, counter electrode, sensitiser/dye and electrolyte. Typically, the conductive substrate is made up of conductive glass materials such as fluorine tin oxide (FTO) glass with one substrate coated with semiconducting materials such as titanium dioxide (TiO<sub>2</sub>) nanoparticles that are sensitised with dyes (ruthenium-based) [3][3]. Meanwhile, the other substrate is coated with counter electrode catalysts, typically platinum or other materials such as MXene  $(Ti_3C_2)$  flakes [4][4] and tungsten disulfide (WS<sub>2</sub>) nanosheets [5][5]. The two electrodes are then joined together with some iodide-triiodide liquid electrolyte is usually deposited in between them to complete the device. Due to the device's simplicity, DSSC have been under intensive research, with the highest efficiency recorded at 14.2% in the year 2020 [6][6], a slight increase from the previous record of 12.3% in 2011 [7][7]. The lack of significant improvement in the efficiency of DSSC highlights their main issue, especially when compared to some other third-generation solar cells, such as perovskite and organic solar cells with-a conversion efficiency of 25.5% and 19%, respectively [8,9][8,9]. Therefore, researchers have been taking a different approach byin reducing the cost of DSSC manufacturing instead while still producing good or comparable efficiency.

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The cost of DSSC manufacturing can be decreased by developing flexible DSSC using flexible materials such as plastic substrate made up of indium tin oxidepolyethyleneterephtalate (ITO-PET) and indium tin oxide-polyethylenenaphtalate (ITO-PEN). These materials are flexible, lightweight, have good transmittance and conductivity while also having with low production cost by utilising roll-to-roll system to mass produce the substrate in the industry [10,11] The ability of these materials to be shaped easily allowed them to be applicable for mobile and wearable applications, encouraging the development of flexible DSSC even further [12]<del>[12]</del>. Despite all the advantages of the flexible plastic based DSSCs, these devices suffer from low conversion efficiency compared to rigid glass-based DSSCs with the highest efficiency is reported at 10.28% [13][13]. A typical rigid glass-based DSSC underwent sintering at high temperature (usually 450°C for TiO2 nanoparticle photoanode) to improve the interparticle connection between the photoanode nanoparticles, remove organic surfactant in the photoanode paste and enhance the adhesion between the photoanode and substrate [14][14]. However, this approach is not applicable for plastic-based DSSC with low thermal stability as the ITO-PET and ITO-PEN polymer substrate would deform at temperature higher than 150°C [11][11]. Hence, a variety of various research has been conducted to enhance the efficiency of DSSC even at low sintering temperature. One of the approaches that have been taken to enhance the efficiency of DSSC is doping the photoanode with metallic cations materials by tuning the band structure of the photoanode

[15][15]. By adding tin (Sn) as dopant to TiO<sub>2</sub> photoanode, Ni et al. [16][16] managed to develop DSSC with good conversion efficiency of 8.75% compared to DSSC with undoped TiO<sub>2</sub> (7.46%). This was attributed to the Sn dopant that expands the electron mobility and raises the flat band potential, eventually improved the performance of DSSC. Dubey et al. [17][17] meanwhile studied a variety of dopants such as vanadium (V), iron (Fe), chromium (Cr) and cobalt (Co) to TiO<sub>2</sub> photoanode, which produced DSSC with efficiency of 3.33%, 4.85%, 3.29% and 3.20% respectively compared to pure TiO<sub>2</sub> (1.31%). Thus, highlighting the effectiveness and potential in using dopants as there are wide availability of materials that can still be studied. Dopants have also shown to be effective for flexible DSSC that was sintered at low temperature of 130°C. This was as demonstrademonstrated by Song et al. [18][18] where magnesium (Mg) was used as doping material to TiO<sub>2</sub> photoanode. An increase in efficiency can be seen as Mg-doped TiO<sub>2</sub> DSSC yielded an efficiency of 2.2% compared to undoped ones (1.7%) due to the change in the conduction band state of TiO<sub>2</sub> surface.

By using the surface plasmon resonance effects found in metallic nanoparticles and the concept of liquid phase sintering, another approach can also be taken to improve the DSSC sintering process. Liquid phase sintering utilises at least two chemical elements mixed at different compositions, with the major component remaining as solid while the minor component becomes the aids that forms liquid during sintering [19]<del>[19]</del>. Liquid phase sintering enhances the sintering mechanism by enhancing the diffusion speed and mass transfer mechanism. Ahmad et al. [20][20] utilises germanium (Ge) as sintering aid to TiO2 photoanode, where they discovered necks were formed at the TiO2-Ge interface during sintering. These necks helped improve the interparticle connection leading to better efficiency from 4.03% for pure TiO<sub>2</sub> photoanode to 5.65% with the addition of Ge sintering aid. In another study, Ahmad et al. [21] applied zinc (Zn) nanoparticles as sintering aid to TiO<sub>2</sub> nanoparticles with lower sintering temperature of 200°C. A similar outcome was obtained with necks forming at the TiO<sub>2</sub>-Zn interfacephase, improving the efficiency from 4.27% for pure TiO<sub>2</sub> photoanode sintered at 450°C, to 4.92%. Due to the low melting point of Zn (419.5°C), the necks were formed despite the low sintering temperature because of the rules of metallurgical sintering. In metallurgical sintering, the sintering temperature of metals are typically found at around 67-75% of the metals' melting point [22][22]. Hence, addition of ing sintering aid for liquid phase is suitable for both high and low temperature DSSC development.

Lead (Pb) has high durability, density and is ais a highly durable\_and corrosion resistant metal with a low melting point of  $327.5^{\circ}$ C [23][23]. Pb is a group IV element similar to Ge and Sn which have shown good photocatalytic activity with <u>TiO2TiO2 which</u> has encouraged some previous studies to try reproduce similar outcomes-as the other metals. Pb was used as dopant material with TiO2 that has tuned the energy bandgap and electronic structure of TiO2, extending the wavelength response for solar light, increasing the number of photogenerated electrons and enhances the photocatalytic activity of Pb-doped TiO2 films [24,25][24,25]. Lead sulphide (PbS) nanoparticles with their tuneable band gap have been used as photoanode material for DSSC in a study by Mishra & Saha [26][26] with power conversion efficiency (PCE) of 1.92%. PbS has also been used with TiO2 photoanode to form nanocomposite paste in a study by Singh et al. [27][27] which produced DSSC with efficiency of 0.35%, compared to the pure TiO2 photoanode (0.08%).

Overall as can be seen from the above literature review, yet extensive studies have not been conducted using <u>pure</u> Pb metal nanoparticles in DSSC development. <u>Besides</u>, the implementation of metal nanoparticles as sintering aid to DSSC photoanode has still not been

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> a widely research topic in this field. However, metal type of nanoparticles have great potential to enhance the power conversion efficiency (PCE) of DSSCs especially for low temperature applications. Thus, this research aims to develop highly efficient DSSC at low sintering temperature with novel implementation of Pb as sintering aid to TiO<sub>2</sub> photoanode. The low melting point and high charge carrier mobility would make them a suitable sintering aid to form necks with the  $TiO_2$  nanoparticles despite the low sintering temperature [23]. Several different compositions of Pb and low sintering temperatures will also be are investigated to study their effects on the performance of DSSC as well as determining the optimal condition. Highest PCE (8.73%) is found to be for specimens containing 5wt% Pb in the TiO<sub>2</sub> matrix at 250°C which outperform as compared to their high-temperature (450°C) counterpart (having PCE of 7.78% ) by 12.21%. Besides, the low cost of Pb due to the abundance of the material is another key factor in supporting the viability of this novel research to develop low cost DSSC with good efficiency. Thus, this research aims to apply Pb nanoparticles as a sintering aid to TiO2 photoanode to develop highly efficiency DSSC even at low sintering temperature. The low melting point of Pb would make them a potentially suitable sintering aid even at low sintering temperature to form necks with TiO2 nanoparticles and eventually improve the efficiency of DSSC. Several different sintering temperatures as well as varying composition of Pb will be investigated in this study to study their effects on the overall performance of DSSC.

#### 2.0 Methodology

#### 2.1 Materials

Titania (TiO<sub>2</sub>) paste (Sigma-Aldrich), fluorine tin oxide (FTO) glass (Sigma-Aldrich, surface resistivity:  $13\Omega/sq$ ), Pb powder (Sigma-Aldrich, 100 mesh), platinum (Pt)-coated FTO glass (Solaronix), iodolyte An-50 (Solaronix), ruthenizer 535-bisTBA (Solaronix), methanol (Fisher Scientific), acetonitrile (Fisher Scientific), ethanol (Fisher Scientific), binder clips.

#### 2.2 Preparation of Pb sintering aid

20g of Pb powder (100 mesh) was deposited into a grinding bowl with zirconia balls for the grinding process using a planetary ball mill <u>(Fritsch Pulverisette 5)</u> operatedran at 500 rpm <del>(Fritsch Pulverisette 5)</del>. Multiple grinding cycles were conducted until the Pb powder was found to be in <u>nanoparticle size of</u> around 450-500 nm, checked by using a particle size analyser (Anton Paar Lifesizer 500). <u>Energy dispersive X-ray (EDX)(EDX model: Oxford Instrument,</u> <u>X-Max65T</u>) was also used for elemental composition analysis to make sure the grinded samples were not contaminated.

#### 2.3 Preparation of photoanode samples

Pure FTO glass was cleaned using methanol each time before they were deposited with photoanode materials. A coating area of 1cm x 1cm was prepared by taping up some parts of the FTO glass using cellophane tape. Pure TiO<sub>2</sub> photoanode samples were prepared by depositing some TiO<sub>2</sub> paste onto the coating area and was coated via doctor blade method. The coated sample was left to dry for 1 hour before the tape was removed, leaving a smooth coated surface. The coated FTO glass was then placed onto a hotplate (IKA RCT basic) and was heated at several different temperature (150°C, 200°C, 250°C and 450°C) for 3 hours. After the sintering process, the samples were then kept and secured for the next steps.

In order to prepare the Pb-TiO<sub>2</sub> photoanode samples, <u>d</u>ifferent amount of grinded Pb powder was added to the TiO<sub>2</sub> paste<u>and mixed</u> to form different weight composition of Pb (0.5wt%, 1wt%, 3wt%, 5wt% and 10wt%). Similar to the previous step, the doctor blade method was used to coat the composite paste onto the FTO glass to form a coating <u>areawith of</u> 1 cm x 1 cm<del>-dimensions</del>. The samples were then heated at different temperatures (150°C, 200°C and 250°C) for 3 hours. The sintered photoanode samples were then kept and secured for further use.

#### 2.4 Fabrication of DSSC devices

Ruthenium-based dye solution was prepared by dissolving 10mg of ruthenizer powder into 5ml of acetonitrile solution. The photoanode sample was soaked into the dye solution and left inside for 1 day. After 24 hours, the photoanode sample was taken out from the dye solution and was left to dry at room temperature for 5 hours. Pt-coated FTO glass was then combined together with the dyed photoanode sample. A drop of electrolyte was deposited into onto-the sample via a small hole on the Pt-coated FTO glass to complete the device. Binder clips were then used to clip the two components together. Similar process was repeated for all the photoanode samples and the devices were then kept for further use. The schematic diagram for the preparation of photoanode samples and DSSC devices was shown in Figure 1.



Figure 1: Schematic diagram of the preparation of photoanode samples and fabrication of DSSC devices

#### 2.5 Characterisation and analysis process

Several characterisation studies were conducted on the photoanode samples using equipment such as the field emission scanning electron microscope (FESEM model: Hitachi SU8010) and transmission electron microscope (TEM model: FEI Tecnai G2 20 S-TWIN) for morphological studies. Elemental composition and mapping of the photoanode samples were also investigated using the scanning electron microscope-energy dispersive x-ray (SEM-EDX model: TESCAN VEGA 3) analysis. Phase analysis was also conducted using x-ray diffraction (XRD) analysis (XRD model: Rigaku Miniflex 600). Light absorption studies were also conducted using UV-visible spectroscopy (UV-vVis model: PerkinElmer Lambda 750). Electrochemical impedance spectroscopy (EIS) analysis was also conducted using Gamry Interface 1010E Potentiostat to plot the Nyquist and Bode plot. Nyquist plot was used to check for the series (Rs) and charge transfer resistance (RCT) while the Bode plot was applied to determine the electron lifetime (t) of the DSSC devices. The frequency range was set at 0.1 Hz - 1 MHz and the voltage were set at 10 mV for all samples. Solar simulator (solar simulator model: IVT 300) was also used (solar simulator model: IVT 300) to measure the currentvoltage (I-V) curves, short circuit current (J<sub>SC</sub>), open circuit voltage (V<sub>OC</sub>), fill factor (FF) and incident photopower conversion efficiency (IPCEPCE). A 1600W Xenon lamp (SAN-EI electric Co., Ltd) was used in The IVT 300 series solar simulator uses a 1600W Xenon lamp (SAN-EI electric Co., Ltd). The caliberation was done before testing using Aa certified golden

solar cell (Amorphous Silicon solar cell) has been used to calibrate the machine before testing. <u>The reason behind using Aa</u> standard Amorphous Si solar cell has been selected as it<u>it was due</u> to showed minimum mismatch and is assumed to be similar in action spectra with DSSCs. The calculated mismatch was found to be approximately 3.7% and, for proper measurements which it was adjusted in machine parameters as a mismatch factor for proper measurements. The data obtained were then analysed and tabulated accordingly in the research. At least 5 samples for each selected sample have been tested and the average value has been presented in this study.

## 3.0 Results and discussion

#### **3.1 Morphological studies**

FESEM equipment have been used to capture the image of the prepared photoanode samples to analyse their morphologies at different sintering temperatures and with the addition of Pb sintering aids. Figure 12 shows the coated and sintered photoanode samples on top of FTO glasses were used for the <u>FESEM</u> analysis and their resultas well as the elemental mapping and composition via the SEM-EDX analysis.<sup>+</sup>



Figure 21; a) FESEM image of pure TiO<sub>2</sub> photoanode sintered at 150°C b) FESEM image of pure TiO<sub>2</sub> photoanode sintered at 250°C c) FESEM image of TiO<sub>2</sub>-5% Pb photoanode sintered at 200°C d) FESEM image of TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C e) Schematic diagram of the necking between the TiO<sub>2</sub>-Pb nanoparticles f)-i) SEM-EDX elemental mapping for Ti, O and Pb elements respectively j) EDX report for TiO<sub>2</sub>-5% Pb photoanode sample FESEM images of prepared photoanode samples a) TiO<sub>2</sub> @ 150°C b) TiO<sub>2</sub> @ 250°C c) TiO<sub>2</sub> = 5%Pb @ 150°C d) TiO<sub>2</sub> = 5%Pb @ 250°C c) Schematic diagram of the TiO<sub>2</sub>-nanoparticles with and without necking

FESEM images of the prepared photoanode samples have been depicted in Figure 12a-d. The effect of sintering temperature can be observed from the FESEM images, where the porosity of the films seems to have increased as the sintering temperature increased. The  $TiO_2$  nanoparticles in Figure 12b have also agglomerated more with a denser arrangement than the nanoparticles in Figure 12a due to the increase in sintering temperature. Thus The increase in sintering temperature has, enlargeding the surface area for better dye absorption, light harvesting and eventually DSSC performance.

Meanwhile, Figures 12c and 12d highlight the presence of Pb sintering aids in the photoanode films. The TiO2 nanoparticles in both figures have shown better connectivity due to the presence of Pb sintering aid as indicated in the figure. The cause for this improved connection can be attributed to the aforementioned liquid phase sintering and metallurgical sintering theory. Pb nanoparticles have been melted during the sintering process due to their low melting point of 327.5°C. Since the sintering temperature of metal nanoparticles is typically around 70% of their melting point, Pb nanoparticles were sintered well at 250°C [22]<sup>[22]</sup>. The melting of Pb nanoparticles then caused the formation of necks at the TiO<sub>2</sub>-Pb interface as shown in Figure 12e, which would enhance the interparticle connectivity of the photoanode. The presence of Pb nanoparticles with the desired elemental composition was also observed in the EDX report of the TiO2-5% Pb photoanode sample,- Elemental mapping for the aforementioned photoanode sample has also been conducted showing the presence of all the elements required in the formation of necks at the TiO<sub>2</sub>-Pb interface. The absence of other elements in the EDX analysis indicates that there were no traces of impurities in the prepared photoanode samples. Hence, highlighting the preparation of photoanode samples have been prepared accordingly in the research...

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Besides FESEM analysis, TEM analysis was also conducted and their images were also observed to study the morphologies of the photoanode samples prepared <u>as</u> shown in Figure



Figure <u>2</u>3: TEM images of photoanode samples <u>a) pure</u> TiO<sub>2</sub> <u>sintered at</u> 250°C <u>b)</u> TiO<sub>2</sub>-5% Pb <u>sintered at</u> 250°C <u>with lower magnification</u> <u>c)</u> TiO<sub>2</sub>-5% Pb <u>sintered at</u> 250°C <u>with</u> <u>higher magnification</u>

Figure 23 showed the TEM images of photoanode samples prepared at 250°C with and without the addition of Pb sintering aid. The effect of the implementation of Pb sintering aid can clearly be seen in Figure 23b & 2c where there has been an increase in the aggregation between the nanoparticles with the formation of larger chunks of nanoparticles. The highlighted section in Figure 23b & 2c indicates the formation of necks between the TiO<sub>2</sub>-Pb nanoparticles during the sintering process. The presence of these nanoparticle aggregations in the TEM images

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further supports the observation made in the FESEM images regarding the formation of necks that improve the interparticle connection.

Phase identification of photoanode samples has been conducted via the XRD analysis as shown in Figure 43.



Figure <u>34</u>: a) XRD analysis of pure TiO<sub>2</sub> photoanode samples at multiple sintering temperature b) XRD analysis of TiO<sub>2</sub> - 5%Pb photoanode samples at multiple sintering temperature with unit cells of anatase TiO<sub>2</sub> and rutile TiO<sub>2</sub> [28][28]

Figure <u>34a</u> <u>indicates</u> that the peaks have been formed at similar areas for each photoanode sample even with different sintering temperatures. This observation <u>indicatesshows</u> that the crystallite structure of TiO<sub>2</sub> did not experience any phase change despite the different sintering temperatures. The more desirable TiO<sub>2</sub> anatase phase have been found to be more prevalent than the less desirable rutile phase as <u>indicated-highlighted</u> in the peaks shown in Figure <u>34a</u>. Peaks were formed at 20 positions of 26°, 39°, 49°, 56° and 64° with miller indices of (101), (004), (200), (211) and (204) planes, respectively, indicating TiO<sub>2</sub> anatase phase [<u>29]</u>[<del>29]</del>. Meanwhile, a few peaks formed at 27° and 36° with miller indices of (110) and (200) planes, respectively which indicated the TiO<sub>2</sub> rutile phase. The unit cells showed that the Ti-Ti distance is longer, while the Ti-O distance is shorter in anatase phase compared to the rutile phase. This structure would then cause a difference in the mass densities and electronic band structures [30][30]. Hence, anatase TiO<sub>2</sub> have larger bandgap, higher surface energy, lower recombination reaction and smaller grain size, for better photocatalytic and photovoltaic activity [29,31][29,31]. These advantages would eventually lead to a higher DSSC efficiency, making TiO<sub>2</sub> anatase phase more favorable. Pb sintering aid has also been found to not have caused a change in the TiO<sub>2</sub> phase. TiO<sub>2</sub> anatase have remained dominant despite adding Pb as indicated by the peaks formed at 32° with miller indices (111) plane as shown in Figure 34b [32][32]. From these observations, it can be deemed that the low sintering temperature of 150-250°C as well as with the addition of Pb nanoparticles as sintering aid were suitable for maintaining good TiO<sub>2</sub> anatase phase.

#### 3.2 Light absorption studies

Light absorption studies for the photoanode materials have been conducted by dissolving an appropriate amount of TiO<sub>2</sub>, Pb, and TiO<sub>2</sub>-5% Pb into ethanol solvent before they were sonicated for 30 minutes. Pure ethanol solution was then used as a reference, and the outcome from the UV-visible spectroscopy was shown in Figure 45. From the UV-vis spectra, the absorbance peak for TiO2 sample was at the wavelength of 350 nm. This falls under the UV region as shown in Figure 4, typical for TiO<sub>2</sub> nanoparticles [33]. Similar finding was also obtained for the TiO2-5% Pb sample where the absorbance peak was formed at a wavelength of 340 nm, also falling in the UV region. The presence of Pb particle can also be found in the nanocomposite with the formation of peak at around 270 nm, similar to the Pb sample. Apart from that, it can be seen that there was higher absorbance for the nanocomposite sample as opposed to pure TiO2 and Pb sample. The addition of Pb nanoparticles was observed to have helped increase the absorbance of TiO2-5% Pb sample even beyond the UV region. This was proven by the absorbance peak has been formed at the near-infrared region with wavelength of 870 nm, similar to the peak formed by the pure Pb nanoparticle as shown in Figure 4. Hence, indicating the impact of adding Pb nanoparticles as sintering aid to add wavelength response in the near-infrared region for the TiO2-Pb nanocomposite sample. The improvement in the nanocomposite sample to absorb light at both UV and near-infrared region would help boost their light absorption capabilities. This would eventually enhance the conversion efficiency of the solar device and encourage the implementation of Pb as sintering aid.

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From the UV-vis spectra, the absorbance peaks for both TiO<sub>2</sub> and TiO<sub>2</sub>-5% Pb was found to have been formed at wavelength of 280 nm. This falls under the UV region as shown in Figure 5, typical for TiO<sub>2</sub> nanoparticles. Higher absorbance has been observed for the nanocomposite as opposed to the pure TiO<sub>2</sub> sample. Thus, indicating the role of Pb in enhancing the absorption of the nanocomposite in the visible light range as highlighted in Figure 5.



Figure 45: UV-visible spectra of Pb, TiO<sub>2</sub> and TiO<sub>2</sub>-5% Pb in ethanol solution

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The difference in the absorbance values between the two samples in the spectra also shows tuning of TiO<sub>2</sub>-energy bandgap due to Pb. Therefore, the wavelength response has been

extended towards the visible light region that would improve the photocatalytic activity of the photoanode [25].

#### 3.3 Electron transfer studies

Figure 56 and 67 depicts the Nyquist and Bode plot from EIS analysis which was performed to study the electron transfer of the complete DSSC devices.

\_EIS response for DSSC devices prepared at 250°C sintering temperature for different Pb composition have been studied. In theory, 250°C was determined to be the most suitable sintering temperature that would yield good results by considering the metallurgical sintering concept\_[22]. The effect of Pb composition was then studied by analysing the EIS response for these DSSC devices.



Figure <u>56</u>: a) Nyquist plot for DSSC devices sintered at 250°C with equivalent circuit b) Bode plot for DSSC devices sintered at 250°C c) Schematic diagram of electron pathway between pure TiO<sub>2</sub> photoanode samples and TiO<sub>2</sub>-Pb composite photoanode sample

From the Nyquist plot, the charge transfer resistance ( $R_{CT}$ ) of DSSC devices was observed to have decreased as Pb sintering aid was implemented to TiO<sub>2</sub> photoanode. The  $R_{CT}$  for DSSC **Formatted:** Caption, Centered, Don't add space between paragraphs of the same style, Line spacing: 1.5 lines

> based on pure TiO<sub>2</sub> photoanode was found to be 22.572  $\Omega/cm^{2_{7}}$  while the R<sub>CT</sub> for DSSC based on TiO<sub>2</sub>-0.5% Pb was 18.317  $\Omega$ /cm<sup>2</sup>. The result indicated that the charge transfer capability of the composite photoanode was better. The improvement was attributed to the neck formation between the TiO2-Pb nanoparticles during the sintering process. The formation of these necks increased the interparticle connectivity from the surface fusion of Pb and the increase in electron density. This occurrence would then led to shorter electron pathway for the transfer of electrons throughout the nanoparticles. With shorter electron pathway, the charge transfer would be improved and the charge transfer properties of the TiO<sub>2</sub>-Pb photoanode has been enhanced., eventualy reduces the electron pathway throughout the nanoparticles. Thus, the recombination reactions and resistance of DSSC devices have been reduced by adding Pb sintering aid as opposed to pure TiO2 photoanode with higher resistance values. The R<sub>CT</sub> was reduced even further as the composition of Pb was increased. The lowest R<sub>CT</sub> value was discovered when TiO<sub>2</sub>-5% Pb photoanode was applied with a value of  $11.54 \Omega/cm^2$ . However, the resistance was found to have increased when the composition of Pb was increased to 10% with high  $R_{CT}$  value of 49.29  $\Omega/cm^2$ , higher than pure TiO<sub>2</sub> photoanode. This occurrence was due to excessive amount of Pb, which became impurities that causes electron trapping sites and surface defects [21][21].

> The results from the Nyquist plot were supported even further with the study of Bode plot. From Bode plot, the electron lifetime (t) of the DSSC devices was determined using Equation (1) [17][33]:

$$t(ms) = \frac{1}{2\pi f_{max}} \tag{1}$$

Where,  $f_{max}$  is the maximum frequency from the Bode plot.

The *t* value for DSSC based on pure TiO<sub>2</sub> photoanode was calculated to be 179.229 ms while the *t* value for DSSC based on TiO<sub>2</sub>-5% Pb photoanode was calculated to be 358.861 ms. The longer the electron lifetime, the lower the possibility of recombination reactions to occur at the TiO<sub>2</sub> network. With low charge transfer resistance, longer electron lifetime and less possibilities of recombination reaction, TiO<sub>2</sub>-5% Pb has been concluded as the most suitable composition for DSSC development. Meanwhile, TiO<sub>2</sub>-10% Pb composition was considered unsuitable and worse than pure TiO<sub>2</sub> due to the high R<sub>CT</sub> and low *t* (25.887 ms), with high recombination reaction probability. Further analysis of DSSC fabricated using the TiO<sub>2</sub>-5% Pb composition was then conducted to study the effect of sintering temperature.

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Figure <u>67</u>: a) Nyquist plot for various DSSC devices with equivalent circuit b) Bode plot for various DSSC devices

EIS analysis for DSSC devices based on TiO<sub>2</sub>-5% Pb composition at different sintering temperature was conducted and compared with pure TiO<sub>2</sub> DSSC devices. Figure <u>67</u>a showed that  $R_{CT}$  values decreased as the sintering temperature increases. The highest  $R_{CT}$  value was found for DSSC based on pure TiO<sub>2</sub> sintered at 150°C with a value of 72.966  $\Omega/cm^2$ . The photoanode's smaller porosity and surface area lowered the electrical contact between nanoparticles with lowand electron transfer that and-increased the resistance and recombination reaction [34][34]. To prevent such issue, TiO<sub>2</sub> photoanode is typically sintered at 450°C [35][35]. As shown in Figure 6a, the pure TiO<sub>2</sub> photoanode sintered at 450°C yielded lower  $R_{CT}$  value of 14.883  $\Omega/cm^2$ , highlighting the importance of having high sintering temperature. However, in this research, the photoanode samples with Pb addition were sintered at low temperature of 150°C-250°C to be applicable for flexible DSSC development [36].

#### TiO2 photoanode is sintered at 250°C for flexible DSSC applications.

Additionally, the absence of sintering aid also hindered the connection between the TiO<sub>2</sub> nanoparticles. Lower  $R_{CT}$  values (22.418  $\Omega/cm^2$ ) was obtained when 5% Pb was added, even at 150°C sintering temperature. The lowest  $R_{CT}$  value of 11.54  $\Omega/cm^2$  was obtained when TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C was implemented. The value was even lower than

the typical control DSSC based on TiO<sub>2</sub> photoanode sintered at 450°C ( $R_{CT}$  = 14.883  $\Omega/cm^2$ ). Due to the concept of liquid phase and metallurgical sintering, Pb, with a melting point of 327.5°C, has melted and formed necks at the TiO<sub>2</sub>-Pb interface [19][19]. The interparticle connection of the photoanode have been greatly enhanced, while the charge transfer resistance and recombination reactions have been lowered. Besides the Nyquist plot, Bode plot have also been analysed to calculate the electron lifetime of DSSC. The longest electron lifetime value was obtained for DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C with a value of 358.861 ms. Thus, lowering the recombination reactions of the DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C has demonstrated the potential to have the best performance.

#### 3.4 I-V curve

The current-voltage (I-V) characteristics of the prepared DSSC devices have been determined using a solar simulator and was plotted in Figure <u>78</u>. From the solar <u>cell</u> I-V curve, several important parameters in evaluating the performance of DSSC have also been calculated and tabulated in Table 1-along with dye loading. Dye loading for the DSSC was also performed <u>via</u> <u>T</u>the absorption-desorption method was performed, and <u>sodium hydroxide (NaOH)</u>-ethanol solution was used to desorb the dye loaded on the photoanode samples. <u>The outcome of the dye loading was also presented in Table 1. Control DSSC</u>\_-device prepared with pure TiO<sub>2</sub>

photoanode sintered at 450°C was used as reference, thus, comparisons can be made between the prepared DSSC and <u>controltypical</u> DSSC.



Figure <u>78</u>: I-V curve for the prepared DSSC devices

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Increased sintering temperature burned out more organic compounds and impurities in  $TiO_2$  nanoparticles, which led to an enlarged surface area [34][34]. This then improved their dye

loading capability, eventually enhancing their light harvesting capability. Meanwhile, improvement in the interparticle connection between the  $TiO_2$  nanoparticles from the addition of Pb sintering aid have also shown to have improved the dye loading ability of the photoanode samples. The highest dye loading was obtained for the  $TiO_2$ -5% Pb photoanode sample sintered at 250°C with a value of 1.17 x  $10^{-7}/cm^2$ , <u>This was</u> higher than the <u>typical control DSSC</u> sample pure  $TiO_2$  photoanode sintered at high temperature of 450°C. Thus, indicating that the prepared nanocomposite photoanode have good dye loading and light absorption capability that would eventually enhance the DSSC performance.

Photoanode sample	Sintering temperature (°C)	(%)	Voc (V)	J <sub>SC</sub> (mA/cm <sup>2</sup> )	FF (%)	Dye loading (× 10 <sup>-7</sup> /cm <sup>2</sup> )
TiO <sub>2</sub>	150	2.70	0.59	6.13	56	1.01
TiO <sub>2</sub>	200	3.87	0.61	9.34	63	1.03
TiO2	250	4.48	0.60	11.14	71	1.03
TiO <sub>2</sub>	450	7.78	0.61	16.93	74	1.06
TiO <sub>2</sub> - 5% Pb	150	4.17	0.60	<u>9.1410.58</u>	71	1.02
TiO <sub>2</sub> - 5% Pb	200	6.86	0.63	13.64	72	1.08
TiO2 - 5% Pb	250	8.73	0.6 <u>4</u> 2	1 <u>7.78</u> 6.93	74	1.17
<u>,TiO2-5%</u> <u>Pb</u>	<u>450</u>	3.03	<u>0.60</u>	7.36	64	<u>n/a</u>

#### Table 1: Summary of parameters of DSSC samples

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 $\label{eq:Voc} Table \ 1 \ shows \ that \ the \ open \ circuit \ voltage \ (V_{OC}) \ values \ \underline{for \ the \ samples \ underwent} \\ \underline{minimal \ changes \ and \ were} \ slightly \ increased \ when \ Pb \ was \ added \ as \ a \ sintering \ aid. \ The \ values \ added \$ 

of  $V_{OC}$  depend on the difference between the Fermi energy level of photoanode and electrolyte's redox potential [37][36]. With the addition of Pb nanoparticles, the conduction band shifted to higher energy and increased the  $V_{OC}$  values. The short circuit current ( $J_{SC}$ ) was discovered to have improved when the sintering temperature was increased and Pb sintering aid was added. The increased  $J_{SC}$  can be attributed to the enhanced dye absorption with higher charge transfer and lower recombination in the photoanode [15,37][15,36]. The increased interparticle contact, interconnected porosity and enlarged surface area from the increase in sintering temperature, have caused these changes, with the highesthigh  $J_{SC}$  value of 16.93 mA/cm<sup>2</sup> obtained for DSSC sintered at 450°C.

Meanwhile, the lowest  $J_{SC}$  (6.13 mA/cm<sup>2</sup>) was yielded by DSSC photoanode sintered at a low temperature of 150°C with a small surface area and low dye absorption. DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at a lower temperature of 250°C has also yielded the highest  $J_{SC}$  value. Thus, indicating that the presence of Pb nanoparticles has also improved the interparticle connectivity, dye absorption, and charge transfer. The recombination reaction has also been lowered as discussed in the EIS analysis. The fill factor (FF) of DSSCs was increased when the charge transfer resistance was reduced. The highest FF value was 74%, yielded by the two DSSC devices with the lowest charge transfer resistance <u>which were the TiO<sub>2</sub>-5% Pb</u> <u>sintered at 250°C and the control DSSC sample sintered at 450°C</u>, as discussed in the EIS analysis.



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#### Figure <u>89</u>: Voc, Jsc, FF and <u>IPCEPCE</u> at different sintering temperatures

Considering the aforementioned parameters' results, the incident photo power conversion efficiency (IPCEPCE) of DSSCs have also been determined. As the V<sub>OC</sub>, J<sub>SC</sub>, and FF values increased, the IPCEPCE value also been increased. The highest IPCEPCE value obtained in the research was 8.73% for DSSC based on TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C. This value was even higher than <u>control</u> DSSC prepared with pure TiO<sub>2</sub> at typically high sintering temperature of 450°C with IPCEPCE of 7.78%. Hence, highlighting the superiority in performance of the TiO<sub>2</sub>-Pb photoanode despite the low sintering temperature. The improved interparticle connection due to the necking at the TiO<sub>2</sub>-Pb interface during the liquid phase sintering process have been attributed as the main cause for such good performance. The yielded IPCEPCE value have also been supported by the low R<sub>CT</sub> value obtained in the EIS analysis. With lower R<sub>CT</sub>, the electrons were easily transferred into the TiO<sub>2</sub> surface, enhancing the DSSC performance [33]. The nanocomposite photoanode have also shown the highest dye loading value of 1.17 x 10<sup>-7</sup>/cm<sup>2</sup>, indicating high light harvesting capability. Besides, DSSC sample based on TiO<sub>2</sub>-5% Pb photoanode sintered at high temperature of 450°C has also been prepared for comparison purposes with the low temperature

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DSSC. The result showed that the PCE value for this sample to be lower at 3.03%. The decrease in efficiency could be attributed to the oxidation of Pb in the TiO<sub>2</sub> matrix since the oxidation temperature of Pb is around 450°C-640°C [38]. Apart from that, the high sintering temperature has caused Pb to be completely melted and causing excessive liquid phase formation that lowers the porosity of the photoanode [39]. Thus, lowering the dye absorption capability of the photoanode and eventually lowers the performance of DSSC. Hence, the DSSC based on TiO<sub>2</sub>-5% Pb @ 250°C photoanode has the best performance that was even better than the typicalcontrol DSSC device. These results in Figure <u>89</u> highlighted the advantage of liquid phase sintering with Pb sintering aid addition method in yielding high performing DSSC even at low sintering temperature.

#### 4.0 Conclusions

The research aims to improve the efficiency of DSSC devices being prepared at relatively lower sintering temperature. Metallic Pb particles as sintering aid has been employed to reduce the sintering temperature of TiO2 based photoanode. Various concentrations of Pb from 0.5-10wt% has been tried at different sintering temperatures to optimise the electrooptical properties. FESEM, SEM-EDX, TEM and XRD have been employed to study the surface morphology, elemental composition and mapping, surface interaction and phase identification respectively. UVv-vis has been used to examine the photo response of the as prepared specimens. Finally, EIS and solar simulator has been employed to study the charge transfer properties and conversion efficiency. Increase in Pb concentration and temperature leads to improved **IPCEPCE**. This is due to improved inter-particle interparticle contact due to surface fusion of Pb. DSSC prepared with TiO<sub>2</sub>-5% Pb photoanode sintered at 250°C exhibited highest **IPCE**PCE value of 8.73% surpassing even the high temperature DSSC control samples by approximately 12.21% prepared with pure TiO<sub>2</sub> photoanode sintered at 450°C (IPCEPCE: 7.78%). The addition of Pb as sintering aid improved the inter-particle interparticle connection leading to increased in current density and a reduction in recombination reactions. Further increase in concentration of Pb did not improve the **IPCEPCE** which may be cor-related with increased impurities in the TiO<sub>2</sub> matrix, which in turn leads to layer cracking and increased charge trapping sites. Besides Pb, other metallic nanoparticles with low melting point should also be researched in the future by using the liquid phase sintering applied in this research. Metal nanoparticles with a lower melting point could produce highly efficient DSSC at lower sintering temperatures that would be suitable for flexible DSSC substrate. The method has shown to be effective for low sintering temperature operation, is a simple process and there is

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wide availability of metal nanoparticles that are potentially suitable to be used for future research.

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#### **Declaration-of-interests**

The authors declare that they have no known competing financial interests or personal relationships that could have influenced the work reported in this paper.

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#### **Declaration of interests**

 $\boxtimes$  The authors declare that they have no known competing financial interests or personal relationships that could have appeared to influence the work reported in this paper.

The authors declare the following financial interests/personal relationships which may be considered as potential competing interests:

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